

Nanobeam photonic crystal cavity light-emitting diodes

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(Received 27 June 2011; accepted 26 July 2011; published online 16 August 2011)

We present results on electrically driven nanobeam photonic crystal cavities formed out of a lateral *p-i-n* junction in gallium arsenide. Despite their small conducting dimensions, nanobeams have robust electrical properties with high current densities possible at low drive powers. Much like their two-dimensional counterparts, the nanobeam cavities exhibit bright electroluminescence at room temperature from embedded 1250 nm InAs quantum dots. A small room temperature differential gain is observed in the cavities with minor beam self-heating suggesting that lasing is possible. These results open the door for efficient electrical control of active nanobeam cavities for diverse nanophotonic applications. © 2011 American Institute of Physics. [doi:10.1063/1.3625432]

Photonic crystal (PC) cavities have attracted much attention in the last decade for their strong light-matter interaction properties. Recently, there has been significant interest in one-dimensional nanobeam PC cavities due to their smaller footprint and exceptional quality (Q) factors, exceeding those of their two-dimensional counterparts.¹ In addition, 1D nanobeams are well suited for coupling light to on-chip waveguides,² as well as for electrostatic tuning.³ Researchers have been able to exploit these properties to develop low threshold lasers,^{4,5} optomechanic crystals,⁶ strongly coupled quantum dot (QD) cavity systems,⁷ as well as chemical sensors.⁸

However, there is a need for efficient electrical injection in active nanobeam devices, which has not yet been addressed. Recently, we demonstrated a world record low threshold PC laser using a lateral *p-i-n* junction formed by ion implantation.⁹ This type of doping layout allows for precise current flow that can be defined lithographically and be applied to arbitrary cavity designs. The previous laser diode was formed out of an L3 PC cavity defect¹⁰ embedded in a 2D membrane and exhibited lasing with a 208 nW threshold at 50 K. In this letter, we show that the same lateral junction process can be applied to 1D nanobeam photonic crystal cavities, yielding room temperature light-emitting diodes (LEDs) at low control powers. In spite of their narrow cross sections, nanobeams can nonetheless direct current efficiently through the sidewalls and deliver charge to the central cavities. The results presented here are a promising step toward practical active nanobeam device architectures.

Electrically contacted nanobeams were fabricated using a similar process flow as in Ref. 9. We start with a 220 nm thick GaAs membrane with three layers of embedded high density (300 dots/ μm^2) InAs quantum dots. Electron beam lithography steps were used to pattern windows in a deposited silicon nitride mask to implant N- (silicon) and P- (beryllium) type dopants. A rapid thermal anneal activated the N and P regions to 6×10^{17} and 2.5×10^{19} cm^{-3} doping densities, respectively, and also blueshifted the peak of the QD ensemble emission from 1310 nm to 1230 nm. E-beam lithography and dry etching next defined the photonic crystal

cavity patterns along with isolation trenches in the GaAs membrane. We chose a common nanobeam cavity design that incorporates a 5-hole linear taper in lattice constant (from $a=322$ to $a=266$) and hole radius ($r=0.22a$) to increase the quality factor.¹¹ The beam widths were 500-600 nm across. Metal contacts were then deposited and the beams were released by removing the sacrificial AlGaAs layer. Figures 1(a) and 1(c) show several scanning electron microscope (SEM) close-up images of a fabricated nanobeam. From the tilted image in Figure 1(b), it is clear that the beams exhibit a small bowing effect likely due to strain from the underlying AlGaAs layer. We do not believe this sagging effect has much of an impact on device performance because of the similarities in results between beams and non-bowing 2D structures. Figure 1(d) shows a finite-difference time domain (FDTD) simulation of the electric field of the cavity design, with a theoretical Q-factor of 95 000.

Electrical measurements were performed at room temperature using a Keithley 2635 sourcemeter with sub-nanoamp resolution and standard electrical probes. Figure 2(a) shows the measured IV characteristics for beams with

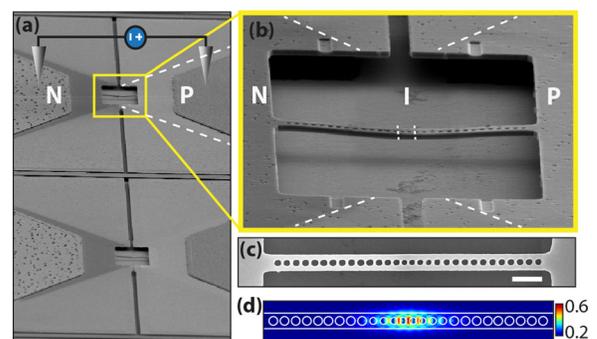


FIG. 1. (Color online) (a) Zoomed out and tilted SEM image of a pair of nanobeam devices. The metal contact pads are seen away from the cavity region and the diagram indicates approximately where electrical probes are placed. The N-type doping is seen as darker grey and the P-type doping is outlined in white dashed lines. (b) Zoomed in SEM of the yellow box region in (a). The beam is deflected down by a small amount likely due to strain from the GaAs/AlGaAs interface. (c) Top view SEM of a nanobeam cavity. The scale bar is 1 μm . (d) FDTD calculated cavity mode electric field magnitude for the designed structure.

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intrinsic region widths of 400 nm and 5 μm as well as for a control sample beam with no holes and an intrinsic region of 400 nm. As expected, the current is substantially reduced by two orders of magnitude when the intrinsic region is large due to the poor diffusion of carriers across the junction. Interestingly, the current magnitudes are very similar for both control and regular samples, with a current near 10 μA for a 1.2 V bias. Despite the reduced current cross sectional area of the beams with holes, the presence of additional etched surfaces creates a greater recombination current. Therefore, the narrow 100-200 nm conducting sidewalls of the beams are large enough for efficient carrier flow. The series resistance of 9000 Ω agrees well with the geometrical sheet resistance of our structure and is correspondingly larger than the 1150 Ω measured in 2D membranes.⁹ We perform 2D Poisson simulations of our devices using the Sentarus package by incorporating measured doping densities, mobilities, and non-radiative recombination lifetime values.¹² Full details of our method will be reported elsewhere. Figure 2(a) displays the simulation current data alongside the experimental data, where reasonable agreement is seen for all three structures. Recombination current accounts for approximately 95% of the total current with the remaining portion due to diffusion. The slightly lower simulation values are likely due to the exclusion of top and bottom device surfaces, for which extra recombination current would be observed. Also, the experimental intrinsic region width is slightly narrower than the simulated width due to dopant diffusion, giving rise to greater current. We find that the diffusion length of electrons (holes) is about 200 nm (40 nm) due to the fast non-radiative recombination in the structure. In Figure 2(b), we see the steady state carrier densities for a 1.2 V bias along the beam length with a 150 nm offset from the center axis. The injected electron and hole densities are only $4 \times 10^{15} \text{ cm}^{-3}$ due to the fast (6 ps) non-radiative recombination lifetime. Two-dimensional plots of the carrier densities for a 1.2 V bias are seen in Figures 2, where it is clear

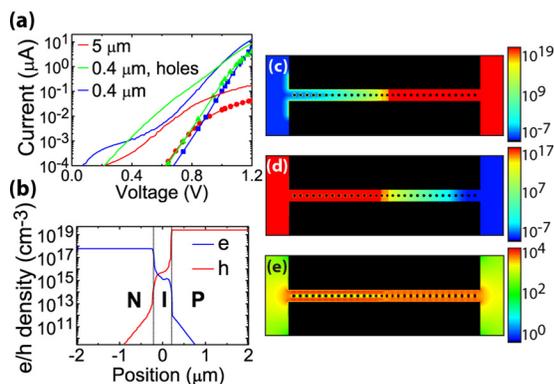


FIG. 2. (Color online) (a) Current-voltage plots of nanobeam structures for a 5 μm intrinsic region with holes, 400 nm intrinsic region with holes, and a 400 nm intrinsic region without holes. Experimental results are given by the solid lines and the simulation results are shown with symbols. (b) Simulated steady-state injected electron (e) and hole (h) density of carriers for a lengthwise cross section slice 150 nm from the central cavity axis for a diode bias of 1.2 V. (c) 2D map of the hole carrier density (in cm^{-3}). Non-radiative recombination was modeled as acceptor-type traps at surfaces and hence there is slight hole accumulation at the edges. (d) 2D map of the electron carrier density (in cm^{-3}). (e) 2D map of the total current density (in A/cm^2).

that the minority carriers are well localized to the cavity region. Finally, Figure 2(e) shows a map of the current density at 1.2 V for which high currents are observed in the beam sidewalls at values up to $10 \text{ kA}/\text{cm}^2$.

Electroluminescence (EL) data was taken at room temperature by forward biasing the nanobeam diodes and collecting the emission with a spectrometer and liquid nitrogen cooled InGaAs CCD array detector. Figure 3(a) displays the output spectrum for a nanobeam biased to 5 μA . Bright cavity mode emission is seen superimposed upon the weaker QD background, indicating successful carrier injection into the nanobeam cavity. The cavity mode has a quality factor of 2900 (Figure 3(b)), well below the theoretical value of 95 000. We believe the quality factor is limited both by fabrication imperfections as well as free carrier absorption by the nearby doping regions. As seen in Figure 3(c), the IR output emission is heavily concentrated to the nanobeam cavity center with a small amount of scattered emission visible at the nanobeam edges. We next investigate the properties of the nanobeams as we increase the injection current. For this experiment, a different cavity than the one seen in Figure 3(a) was used due to accidental device failure. The quality factor of the mode probed in this cavity is about 500. In Figure 3(d), the cavity output emission and Q-factor are plotted versus injection current. The cavity power output is linear for the entire range and therefore no lasing is observed. A small amount of room temperature linewidth narrowing was observed as the Q-factor increased with injection current. Therefore, we believe it is possible to obtain lasing in these important nanophotonic structures with much higher Q cavities.¹³

In order to characterize the heating effects of electrically pumped nanobeams, we examine the mode peak wavelength as a function of injection current. Figure 4(a) shows that the mode wavelength shifts by less than 1 nm for 5 μA of injection current. The thermal dependence of the refractive index of GaAs near 1.3 μm is given by $dn = 2.7 \times 10^{-4} dT \text{ K}^{-1}$, where dT is the change in temperature and dn is the change

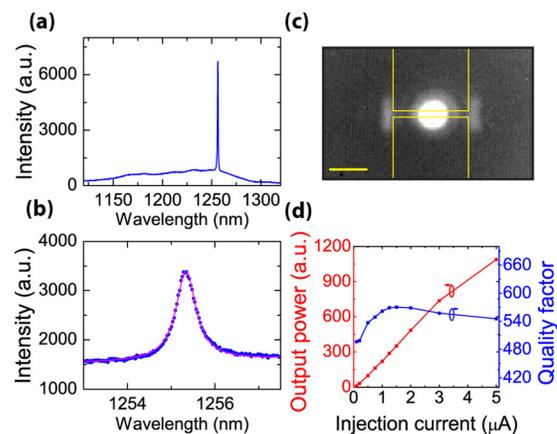


FIG. 3. (Color online) (a) EL spectrum for a nanobeam device at a forward bias of 5 μA . The cavity fundamental mode is the sharp peak at 1255 nm and the background QD emission is the broad spectrum below. (b) A zoom-in spectrum of the cavity peak in (a) along with a Lorentzian fit, giving a Q-factor of 2900. (c) IR camera picture of the nanobeam cavity emission. An outline of the cavity is seen by the lines and the scale bar is 5 μm . The cavity emission is bright at the center as expected and there is slight EL scattered out at the nanobeam edges. (d) Plot of the cavity output power and Q-factor versus injection current.

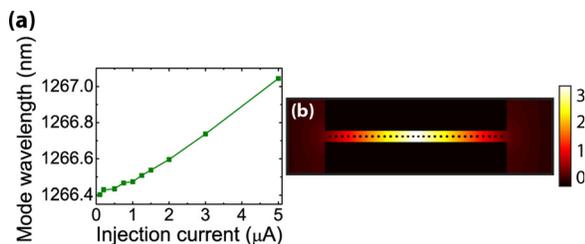


FIG. 4. (Color online) (a) Plot of the cavity peak wavelength versus injection current. (b) 2D map of the change in temperature (in ΔK) of the nano-beam device for a bias voltage of 1.2 V. The cavity center only heats by about 3.3 K.

in material refractive index.¹⁴ Hence the cavity wavelength is expected to shift via second order perturbation theory as $d\lambda/\lambda = dn/n$, where $d\lambda$ is the change in mode wavelength, λ is the cavity peak wavelength, and n is the nominal material refractive index. For a measured $d\lambda$ of 0.6 nm, $\lambda = 1266$ nm, and $n = 3.5$, the calculated temperature rise is only 6 K. For comparison, we calculate the lattice temperature from a hydrodynamic transport model in Sentaurus and find that the heating is 3.3 K at the cavity center. Therefore, our electrical design is very robust against self-heating despite the large injection current densities.

In summary, we have demonstrated efficient electrically driven photonic crystal nanobeam cavity LEDs at room temperature. The results here are an extension of our lateral p - i - n junction design in 2D PCs showing the versatility of the fabrication technique. Our nanobeams have excellent electrical properties even with their narrow conducting paths. Future designs could incorporate surface passivation techniques to slow down non-radiative recombination and hence increase the charge injection level. Other geometrical modifications such as beam width and hole size could further optimize the device performance. The electrical control of nanobeam cavities demonstrated here is an important step forward in developing practical on-chip devices for diverse applications such as lasers, sensors, and optomechanics.

Gary Shambat and Bryan Ellis were supported by the Stanford Graduate Fellowship. Gary Shambat is also supported by the NSF GRFP. The authors acknowledge the financial support of the Interconnect Focus Center, one of the six research centers funded under the Focus Center Research Program, a Semiconductor Research Corporation program. We also acknowledge the AFOSR MURI for Complex and Robust On-chip Nanophotonics (Dr. Gernot Pomrenke), grant number FA9550-09-1-0704, and the Director, Office of Science, Office of Basic Energy Sciences, Materials Sciences and Engineering Division, of the U.S. Department of Energy under Contract No. DE-AC02-05CH11231. Work was performed in part at the Stanford Nanofabrication Facility of NNIN supported by the National Science Foundation. We also acknowledge Kelley Rivoire for assisting in SEM image acquisition.

¹M. Notomi, E. Kuramochi, and H. Taniyama, *Opt. Express* **16**, 11095 (2008).

²Q. Quan, P. B. Deotare, and M. Loncar, *Appl. Phys. Lett.* **96**, 203102 (2010).

³I. W. Frank, P. B. Deotare, M. W. McCutcheon, and M. Loncar, *Opt. Express* **18**, 8705 (2010).

⁴Y. Zhang, M. Khan, Y. Huang, J. H. Ryou, P. B. Deotare, R. Dupuis, and M. Loncar, *Appl. Phys. Lett.* **97**, 051104 (2010).

⁵Y. Gong, B. Ellis, G. Shambat, T. Sarmiento, J. S. Harris, and J. Vuckovic, *Opt. Express* **18**, 5964 (2010).

⁶M. Eichenfield, J. Chan, R. Camacho, K. J. Vahala, and O. Painter, *Nature (London)* **462**, 78 (2009).

⁷R. Ohta, Y. Ota, M. Nomura, N. Kumagai, S. Ishida, S. Iwamoto, and Y. Arakawa, *Appl. Phys. Lett.* **98**, 173104 (2011).

⁸B. Wang, M. A. Dunder, R. Notzel, F. Karouta, S. He, and R. W. van der Heijden, *Appl. Phys. Lett.* **97**, 151105 (2010).

⁹B. Ellis, M. A. Mayer, G. Shambat, T. Sarmiento, J. Harris, E. E. Haller, and J. Vučković, *Nature Photon.* **5**, 297 (2011).

¹⁰Y. Akahane, T. Asano, B. S. Song, and S. Noda, *Nature* **425**, 944 (2003).

¹¹P. B. Deotare, M. W. McCutcheon, I. W. Frank, M. Kahn, and M. Loncar, *Appl. Phys. Lett.* **94**, 121106 (2009).

¹²G. Shambat, B. Ellis, M. Mayer, A. Majumdar, E. E. Haller, and J. Vučković, *Opt. Express* **19**, 7530 (2011).

¹³M. Nomura, S. Iwamoto, K. Watanabe, N. Kumagai, Y. Nakata, S. Ishida, and Y. Arakawa, *Opt. Express* **14**, 6308 (2006).

¹⁴J. Talghader and J. S. Smith, *Appl. Phys. Lett.* **66**, 335 (1995).